



Probing Polyelectrolyte Adsorption in Charged Nanochannels by Streaming Potential Measurements

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ABSTRACT: It remains a great experimental challenge to obtain quantitative information on the polyelectrolyte (PE) behavior confined in charged nanoporous materials. Here, we propose an original approach using transverse streaming potential measurements (TSPMs), an efficient technique providing information on the electrical surface properties of nanoporous materials through the ζ -potential determination. We conduct TSPMs within the thin double-layer approximation on a model system composed of individual nanochannels, a nanoporous anodic aluminum oxide (AAO) membrane, filled with a well-known PE, sodium polystyrenesulfonate (NaPSS). We demonstrate that TSPMs can provide the AAO ζ -potential under different experimental conditions and monitor the PE penetration in AAO with positive or negative surface charge. On the positive surface, the PE irreversibly adsorbs, while it does not when the surface is negatively charged, indicating the electrostatic nature of the PE adsorption. In the context of



experimental limitations to investigate PE behavior on concave surfaces, this study shows that the TSPM is suitable to extract quantitative information and can be exploited to gain an understanding of the PE adsorption and desorption in a confined medium.

P olyelectrolytes (PEs) are polymers carrying electric charges due to the dissociation in polar solvents of ionic groups present on the chain.¹ The charges have direct consequences on the PE behavior in solution compared to neutral polymers.² When a PE is confined in nanoporous materials, its properties (conformation, dynamics, transport, or adsorption) are affected by the geometrical or electrostatic constraints imposed by the confining medium.³ The better understanding of these modifications is relevant for many potential applications such as ionic separation,⁴ catalysis,⁵ or ionic rectification for singlemolecule analysis.⁶ However, few experimental studies^{7–11} investigated the effects of such confinement on the PE characteristics, presumably due to experimental difficulties to collect quantitative information.

In particular, when dealing with charged interfaces, the surface charge of the nanoporous material plays a critical role, and its determination is crucial for further experimental investigations involving PEs. The classical techniques used to characterize surface charges of nanoparticles in solution (such as electrophoretic light scattering¹² or acid/base titration¹³) cannot be applied for porous materials, and alternative approaches have to be found. For instance, indirect information on the charge can be obtained by the titration of radioactive elements after diffusion and adsorption in the porous medium: if positive ions adsorb, then the surface is negatively charged.¹⁴ However, to have more quantitative information on the charge, experiments based on electrokinetic phenomena such as electro-osmosis or streaming potential measurements (SPMs) to determine the ζ -potential have to be used.^{15–19} However, so far, a very limited number of

experimental studies really dealt with such characterization, bringing out the challenging nature of experiments in nano-porous media.

Here, we address these issues by presenting an original approach using SPMs on a model nanoporous system, namely, nanoporous anodic aluminum oxide (AAO) membranes, composed of parallel and nonconnected nanochannels. We show that a SPM is an efficient technique to access the AAO electrical surface properties under different experimental conditions and to further monitor the PE diffusion in AAOs with positive or negative surface charge.

The AAO membranes are prepared by the classical two-step electrochemical anodization of high purity aluminum foil (Al, 99.999% from Goodfellow, 320 μ m thick). A first anodization is performed in 0.3 M oxalic acid solution constantly stirred under a continuously applied voltage of 40 V at T = 18 °C. After 2 h, both sides of the Al foil present the formed oxide, which is subsequently dissolved in an aqueous solution of phosphochromic acid (6 wt % of H₃PO₄ and 1.8 wt % of CrO₃) during 2 h at T = 50 °C. Then, a second anodization is performed for 8 h under the same conditions as the first one. Finally, the AAO membrane

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is obtained by the detachment of the oxide from the remaining Al through a third anodization step made in 12 M H_2SO_4 under 40 V and T = 0 °C for 7 min followed by an immersion in phosphochromic acid during 30 min at 30 °C.²⁰ During this last step, the alumina is easily detached from both sides of the Al substrate, and two through-hole membranes are obtained.

Figure 1 shows scanning electron microscopy images (SEM, using a Hitachi SU-70 FEGSEM under an accelerating voltage of



Figure 1. SEM image of a detached AAO membrane prepared in 0.3 M oxalic acid under 40 V: (a) top surface, (b) bottom surface, and (c) section view. The AAO total thickness is around $62 \,\mu$ m. Scale bar: 1 μ m.

1 or 3 kV) for a typical AAO membrane prepared under the conditions detailed above. From the image analysis, one obtains a pore diameter D_p of 40 \pm 1 nm for the top surface (Figure 1a), while the bottom D_p was systematically larger with a value of 66 \pm 3 nm (Figure 1b). This difference is due to the high dissolution of alumina produced in 12 M H₂SO₄ during the third anodization that enlarges the bottom pore. SEM images from the section view show that the nanochannels are straight and not interconnected (Figure 1c) and adopt a bilayer structure composed of a thick layer of \approx 58 μ m with a D_p of 42 nm and a thin layer of $\approx 4 \,\mu m$ with a D_p of 66 nm (see Figure S1). This residual thin layer comes from the third anodization and has a negligible contribution to the TSPM,²¹ also confirmed by complementary water flow measurements (see details in the SI) through the AAO,²² giving a hydrodynamic diameter of 47 nm, close to the one of the top surface D_p . Finally, since the aspect ratio is greater than 1000, the AAOs can be considered as a collection of infinitely long nanochannels. Note that, for this

work, we synthesized several AAOs under the same conditions to perform all SPMs, and the average $D_{\rm p}$ values are 38 ± 5 nm and 62 ± 4 nm for the top and bottom surface, respectively, indicating the good reproducibility of the AAO synthesis.

Then, SPMs have been performed to quantify the AAO electrical surface properties. In contact with water, the pore-wall surface of aluminum oxide undergoes chemical reactions producing metal-hydroxyl groups whose protonation and deprotonation are responsible for the charge of the surface. Because of the electroneutrality, these static charges are compensated by an imbalance in the amount of anions and cations in solution. Thus, the flow of an electrolyte solution through an AAO membrane caused by a difference of pressure Pon its two sides produces an electric current (namely, the streaming current) that comes with a difference of electric potential between the two sides (namely, the streaming potential $(U_{\rm str})^{23}$ The measurement of the variation of $U_{\rm str}$ with the pressure difference gives direct access to the ζ -potential, which is defined as the electric potential at a distance from the pore-wall surface at which the liquid flow velocity is zero and is given by the following expression

$$\zeta = \frac{\sigma \eta}{\varepsilon} \frac{dU_{\rm str}}{dP} \tag{1}$$

where σ is the electrolyte conductivity; η is the viscosity; and ε is the dielectric permittivity. Equation 1 is obtained by neglecting the contribution of surface conductivity of the electric double layer to the pore conductance (for discussion and details on this point see the SI). This assumption is also confirmed by the literature where the surface conductance is typically neglected for $\kappa R_p > 10$ and low zeta potential value²⁴ (here we use as electrolyte 50 mM KCl, with a Debye length κ^{-1} of 1.4 nm at 298 K and a $\kappa R_p \approx 11.5$, indicating the nonoverlapping of the Debye length).

Previous SPMs on AAOs have been performed by imposing a flow along the surface of the membrane (tangential SPM)¹⁸ or through the membrane (transverse SPM or TSPM),^{16,17,25} the latter being only used so far for large AAO pore sizes (around 0.1 and 0.2 μ m). Although the TSPM is mainly sensitive to the internal nanochannel surface, which is by far the largest, it has not been used to probe the PE behavior inside nanochannels, an approach we want to address. Here, TSPMs are performed with a homemade device consisting of two compartments separated by the AAO membrane and containing the electrolyte and one Ag/Ag-Cl electrode each. Electric measurements are carried out with the electrophysiology amplifier Axopatch 200B used in "I = 0" mode. The pressure difference is measured with a piezoelectric differential sensor and obtained by means of air compressed by a syringe pump applied to one of the compartments, while the other is left at atmospheric pressure. Different buffer solutions with pH ranging from 3.6 to 10.7 (listed in Table S3) were prepared and diluted with ultrapure water to reach a concentration of the buffer of 2 mM. KCl was then added at a fixed concentration of 50 mM to set the solution's conductivity to a constant value of 0.68 S/m. Note that the AAO stability in water has been checked with time, and the ζ -potential value remains constant over a period of 16 h (time larger than our longest measurement time), indicating that the possible AAO hydration has no influence on the ζ -potential (Figure S5).

Figure 2 shows the evolution of the AAO ζ -potential as a function of pH. In acidic conditions (pH < 5.5), the ζ -potential is positive, while it is negative in basic conditions (pH > 9) with a



Figure 2. ζ -potential of AAO membranes as a function of the pH in different buffer solutions (the N° correspond to the N° listed in Table S1). The reported values are obtained at equilibrium, and the error bars correspond to the standard deviation of different measurements done on different samples.

sign inversion (or isoelectric point IEP) around a pH of 8.5. On the contrary, in neutral conditions (5.5 < pH < 8), the sign of the ζ -potential depends on the nature of the buffer molecules: the value is always positive (acetic acid/acetate (N° 1 to 5), ammonia/ammonium (N° 7), or KOH (N° 8)), except for the phosphate buffers where it is negative (N° 15 to 18).

The negative ζ -potential obtained when tuning the pH with phosphate buffers must be attributed to specific adsorption of phosphate ions and more specifically to divalent HPO₄²⁻, which is the dominant species in this pH region (pK_{a1} = 2.12, pK_{a2} = 7.2). Indeed, divalent and trivalent ions have been reported to adsorb strongly on the surfaces and shift the IEP of charged membranes.¹⁹ Moreover, the phosphate adsorption has a slow kinetics as shown in Figure S4a (around 15 min to reach a constant negative value of -10 mV), and phosphate ions can be desorbed by washing the AAO with acetic acid/acetate buffer at pH = 4.6 (see Figure S4b).

From the literature, if there is no specific ion adsorption, the point of zero charge (PZC) or the isoelectric point (IEP) varies from pH = 7.5 to 9.5 for alumina nanoparticles (NPs)²⁶ and from pH = 4 to 7 for a flat surface.²⁷ This wide range of PZC/IEP is mainly due to variations in the hydroxide coordination that depends on the crystal structure and composition.²⁷ Here, our measured IEP (pH = 8.5) is similar to NPs and to previous AAO membranes prepared in 0.22 M oxalic acid for which the IEP measured by tangential SPMs varies from pH = 7.8 to 6.7 when the pore diameter decreases from 40 to 15 nm.¹⁸

Finally, this ion-specific adsorption is a very convenient way to tune negatively the AAO charges to avoid working under basic conditions (where fast AAO dissolution occurs) or with grafting molecules whose charge is pH dependent. This strategy will be exploited in the following to study the PE behavior in AAO with positive or negative surface charge under neutral pH conditions.

We used the widely studied sodium polystyrenesulfonate¹ as the PE (NaPSS purchased from Sigma-Aldrich, $M_w = 70$ kDa, PDI = 1.8), which is a strong PE carrying one charged group per monomer. In situ TSPMs have been conducted in such conditions that the PE diffusion is favored, i.e., $D_H < D_P$ (where $D_H = 20$ nm is the hydrodynamic PE diameter) and working at low PE concentration C_{PE} , i.e., in the dilute regime $(C_{\text{PE}} < C^*$, where the overlap concentration C^* is estimated of the order of 26 g/L from specific viscosity measurements). As mentioned above, we work at neutral pH (pH = 6.5) where the AAO surface can be positive in pure water or negative in phosphate buffer solution. Note that, as for TSPMs in empty AAOs, KCl is added in the PE solution at a concentration of 50 mM to keep the conductivity at a constant value.

We will start with the case of positive AAO. Figure 3 shows the ζ -potential evolution with time of an AAO immersed in NaPSS



Figure 3. ζ -potential evolution with time for AAOs measured in NaPSS aqueous solution (pure water at pH = 6.5) at *C* = 0.14 g/L (black circles) followed by a washing in acetic acid/acetate buffer solution at pH = 3 (red squares). The red continuous line corresponds to the best fit with a simple exponential.

aqueous solution at a concentration of 0.14 g/L. At t = 0, the AAO is positively charged (ζ -potential around 25 mV) and gradually reverses from positive to negative, until the ζ -potential stabilizes at a constant value of -23 mV, indicating the gradual adsorption of PE chains at the surface until the equilibrium is reached. This constant ζ -potential evolution over a long time (more than 1000 min) is consistent with previous works indicating that, at low salt concentration (<0.1 M), the first adsorbed PE chains remain strongly adsorbed and are not displaced by other chains,²⁸ suggesting that there is no chain reorganization with time. The time evolution of the ζ -potential can be fitted with a simple exponential to extract a characteristic time of adsorption $\tau_{\rm C}$ of around 300 min, suggesting a slow adsorption process. The fitting with one simple exponential supposes that PE adsorption on the external top and bottom AAO surfaces does not influence the PE entering process into the nanochannels. Immediately after the PE adsorption, in order to evaluate the strength of the PE adsorption, the TSPM cell was filled with acetic acid/acetate buffer (pH = 3), and the ζ potential is measured over a given time (red squares in Figure 3). The ζ -potential remains negative around -15 mV, indicating that the adsorbed PE chains are not washed away. The increase from -23 mV to -15 mV suggests either some desorption of loosely adsorbed chains or modification of the conformation of the adsorbed layer. Note that the adsorbed chains cannot be removed after a sequence of washing in neutral/basic/acid/ neutral buffers, indicating the irreversibility of the PE adsorption (see Figure S6). The hydrodynamic PE layer thickness is estimated to be around 1 nm by flow measurements (see Figure S2). The PSS-covered nanochannel surface can still be approximated as an impermeable hard surface from which eq

1 is still valid. For a thicker soft permeable layer, appropriate electrohydrodynamic theories have to be used to correctly interpret streaming potential measurements.^{29,30}

We will now focus on the case of negative AAO. For direct comparison with a positive surface, we work under the same neutral pH condition. Figure 4 presents the AAO ζ -potential



Figure 4. AAO ζ -potential evolution with time throughout the following sequence of immersions: water at pH = 6.5 (black squares), phosphate buffer at pH = 6.1 (orange hexagons), NaPSS at 1.4 g/L in phosphate buffer solution at pH = 6.1 (black circles), and acetic acid/ acetate at pH = 4 (red circles).

evolution with time according to a given sequence of buffer solutions. First, in region I, the AAO is immersed in pure water (pH = 6.5), and the ζ -potential value is positive, as expected, and around +15 mV. Then, we immersed AAO in a phosphate buffer solution at pH = 6.1 (region II), and the ζ -potential was found to reverse to a negative value (around -15 mV). As mentioned previously, phosphate ions adsorb at the surface, and the AAO becomes negatively charged. In region III, the negatively charged AAO is immersed in a NaPSS solution prepared in phosphate buffer at pH = 6.1 (C_{PE} = 1.4 g/L). The ζ -potential remains unchanged (around -15 mV), but we still cannot conclude whether PE adsorption took place or not. The PE might be competing with phosphate ions for adsorption. Finally, the AAO is immersed in a acetic/acetate solution at pH = 4(region IV). If the PE chain was adsorbed, the ζ -potential would not vary (as in Figure 3). On the contrary, if the ζ -potential reversed to a positive value, then only phosphate ions would be adsorbed at the surface. One can see that the ζ -potential values came back to +15 mV, indicating that the PE was not adsorbed and the phosphate ions were washed away from the surface. By combining the results obtained for positive and negative AAOs, we can conclude that the NaPSS adsorption is mainly electrostatic (occurs only on positive surfaces) and is irreversible.

The experimental determination of the electrical surface properties of nanoporous materials is challenging. Here we show that TSPM is an efficient technique to:

 (i) determine the ζ-potential of a model porous system (AAO) where the surface charge can be tuned to positive or negative values depending on the nature of the buffer solution; (ii) follow the ζ -potential evolution when a PE diffuses inside. We show that, on a positive surface, the PE adsorbs irreversibly, while it does not when the surface is negatively charged, indicating the electrostatic nature of the PE adsorption.

In the context of experimental limitations to investigate PE behavior on concave surfaces, which are expected to differ from convex ones according to recent theoretical work,³¹ this study shows that TSPMs are suitable to extract quantitative information regarding PE adsorption. It should further enable us to highlight the role of several parameters such as nanochannel size, ionic strength, PE nature, molecular weight, and concentration in the adsorption and desorption processes.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsmacrolett.0c00172.

Experimental details (SEM section view, flow measurements, and TSPMs) and complementary ζ -potential measurements (PDF)

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Notes

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REFERENCES

 Muthukumar, M. 50th anniversary perspective: A perspective on polyelectrolyte solutions. *Macromolecules* 2017, 50 (24), 9528–9560.
 Dobrynin, A. V.; Colby, R. H.; Rubinstein, M. Scaling theory of polyelectrolyte solutions. *Macromolecules* 1995, 28 (6), 1859–1871. (3) Park, P. J.; Chun, M.-S.; Kim, J.-J. Partitioning and conformational behavior of polyelectrolytes confined in a cylindrical pore. *Macromolecules* **2000**, *33* (23), 8850–8857.

(4) Armstrong, J. A.; Bernal, E. E. L. n.; Yaroshchuk, A.; Bruening, M. L. Separation of ions using polyelectrolyte-modified nanoporous tracketched membranes. *Langmuir* **2013**, *29* (32), 10287–10296.

(5) Dotzauer, D. M.; Dai, J.; Sun, L.; Bruening, M. L. Catalytic Membranes Prepared Using Layer-by-Layer Adsorption of Polyelec-trolyte/Metal Nanoparticle Films in Porous Supports. *Nano Lett.* **2006**, *6* (10), 2268–2272.

(6) Ali, M.; Yameen, B.; Cervera, J.; Ramírez, P.; Neumann, R.; Ensinger, W.; Knoll, W.; Azzaroni, O. Layer-by-Layer Assembly of Polyelectrolytes into Ionic Current Rectifying Solid-State Nanopores: Insights from Theory and Experiment. J. Am. Chem. Soc. **2010**, 132 (24), 8338–8348.

(7) Gilbert, E. P.; Auvray, L.; Lal, J. Structure of polyelectrolyte chains confined in nanoporous glass. *Macromolecules* **2001**, *34* (14), 4942–4948.

(8) Oukhaled, G.; Bacri, L.; Mathe, J.; Pelta, J.; Auvray, L. Effect of screening on the transport of polyelectrolytes through nanopores. *Epl* **2008**, *82* (4), 48003.

(9) Deen, W. M.; Smith, F. G. Hindered diffusion of synthetic polyelectrolytes in charged microporous membranes. *J. Membr. Sci.* **1982**, *12* (2), 217–237.

(10) Lazzara, T. D.; Lau, K. H. A.; Abou-Kandil, A. I.; Caminade, A.-M.; Majoral, J.-P.; Knoll, W. Polyelectrolyte Layer-by-Layer Deposition in Cylindrical Nanopores. *ACS Nano* **2010**, *4* (7), 3909–3920.

(11) Christoulaki, A.; Chennevière, A.; Dubois, E.; Jouault, N. Duplex nanoporous alumina and polyelectrolyte adsorption: more insights from a combined neutron reflectivity and electron microscopy study. *Nanoscale* **2019**, *11* (5), 2148–2152.

(12) Doane, T. L.; Chuang, C. H.; Hill, R. J.; Burda, C. Nanoparticle zeta-potentials. *Acc. Chem. Res.* **2012**, *45* (3), 317–326.

(13) Lucas, I. T.; Durand-Vidal, S.; Dubois, E.; Chevalet, J.; Turq, P. Surface Charge Density of Maghemite Nanoparticles: Role of Electrostatics in the Proton Exchange. *J. Phys. Chem. C* 2007, *111* (50), 18568–18576.

(14) Bluhm, E. A.; Bauer, E.; Chamberlin, R. M.; Abney, K. D.; Young, J. S.; Jarvinen, G. D. Surface Effects on Cation Transport across Porous Alumina Membranes. *Langmuir* **1999**, *15* (25), 8668–8672.

(15) Nyström, M.; Lindström, M.; Matthiasson, E. Streaming potential as a tool in the characterization of ultrafiltration membranes. *Colloids Surf.* **1989**, *36* (3), 297–312.

(16) Hernandez, A.; Martinez, F.; Martín, A.; Prádanos, P. Porous Structure and Surface Charge Density on the Walls of Microporous Alumina Membranes. J. Colloid Interface Sci. **1995**, 173 (2), 284–296.

(17) Winkler, B. H.; Baltus, R. E. Modification of the surface characteristics of anodic alumina membranes using sol-gel precursor chemistry. *J. Membr. Sci.* **2003**, *226* (1), 75–84.

(18) Pedimonte, B. J.; Moest, T.; Luxbacher, T.; von Wilmowsky, C.; Fey, T.; Schlegel, K. A.; Greil, P. Morphological zeta-potential variation of nanoporous anodic alumina layers and cell adherence. *Acta Biomater.* **2014**, *10* (2), 968–974.

(19) Burns, D. B.; Zydney, A. L. Buffer effects on the zeta potential of ultrafiltration membranes. J. Membr. Sci. 2000, 172 (1–2), 39–48.

(20) Yanagishita, T.; Masuda, H. High-Throughput Fabrication Process for Highly Ordered Through-Hole Porous Alumina Membranes Using Two-Layer Anodization. *Electrochim. Acta* **2015**, *184*, 80– 85.

(21) Szymczyk, A.; Labbez, C.; Fievet, P.; Aoubiza, B.; Simon, C. Streaming potential through multilayer membranes. *AIChE J.* **2001**, *47* (10), 2349–2358.

(22) Kostaras, C.; Dellis, S.; Christoulaki, A.; Anastassopoulos, D. L.; Spiliopoulos, N.; Vradis, A.; Toprakcioglu, C.; Priftis, G. D. Flow through polydisperse pores in an anodic alumina membrane: A new method to measure the mean pore diameter. *J. Appl. Phys.* **2018**, *124* (20), 204307.

(23) Delgado, Á. V. Interfacial electrokinetics and electrophoresis; CRC Press: 2001; Vol. 106.

(24) Delgado, A. V.; González-Caballero, F.; Hunter, R. J.; Koopal, L. K.; Lyklema, J. Measurement and interpretation of electrokinetic phenomena. *J. Colloid Interface Sci.* **2007**, *309* (2), 194–224.

(25) Szymczyk, A.; Fievet, P.; Aoubiza, B.; Simon, C.; Pagetti, J. An application of the space charge model to the electrolyte conductivity inside a charged microporous membrane. *J. Membr. Sci.* **1999**, *161* (1), 275–285.

(26) Kosmulski, M. Chemical properties of material surfaces; CRC press: 2001.

(27) Franks, G. V.; Gan, Y. Charging behavior at the alumina-water interface and implications for ceramic processing. *J. Am. Ceram. Soc.* **2007**, *90* (11), 3373–3388.

(28) Sukhishvili, S. A.; Granick, S. Kinetic regimes of polyelectrolyte exchange between the adsorbed state and free solution. *J. Chem. Phys.* **1998**, *109* (16), 6869–6878.

(29) Duval, J. F. L.; Zimmermann, R.; Cordeiro, A. L.; Rein, N.; Werner, C. Electrokinetics of Diffuse Soft Interfaces. IV. Analysis of Streaming Current Measurements at Thermoresponsive Thin Films. *Langmuir* **2009**, *25* (18), 10691–10703.

(30) Zimmermann, R.; Dukhin, S. S.; Werner, C.; Duval, J. F. L. On the use of electrokinetics for unraveling charging and structure of soft planar polymer films. *Curr. Opin. Colloid Interface Sci.* **2013**, *18* (2), 83–92.

(31) Gilles, F. M.; Boubeta, F. M.; Azzaroni, O.; Szleifer, I.; Tagliazucchi, M. Modulation of Polyelectrolyte Adsorption on Nanoparticles and Nanochannels by Surface Curvature. *J. Phys. Chem. C* 2018, 122 (12), 6669–6677.